

**ACTIVE, Si-BASED PHOTONIC BANDGAP AND  
MICROPHOTONIC STRUCTURES BASED ON  
RARE EARTH DOPED HYDROGENATED  
AMORPHOUS Si ALLOYED WITH CARBON**

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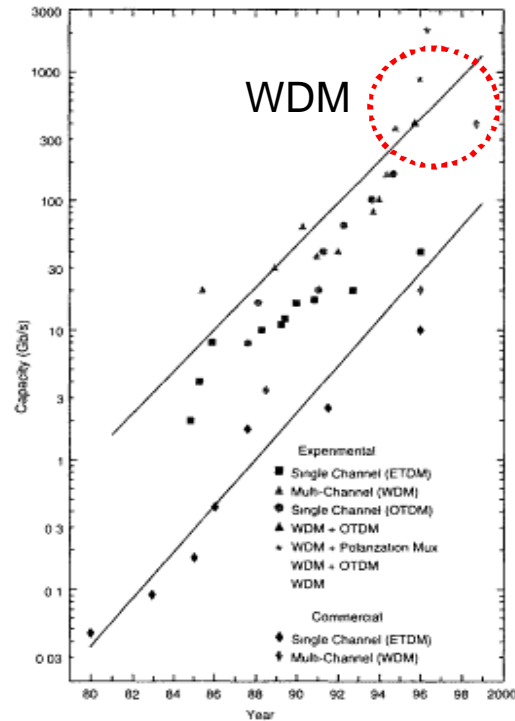
Daejeon, Korea

# Outline

- **Si based microphotronics**
- **a-Si:H:C as a viable material for Si-based microphotronics**
- **Er-doped a-Si:H:C**
  - **Structural properties**
  - **Optical properties**
- **Microphotronics using a-Si:H:C**
  - **Photonic crystal**
  - **Microdisk**
- **Conclusion**

# Introduction

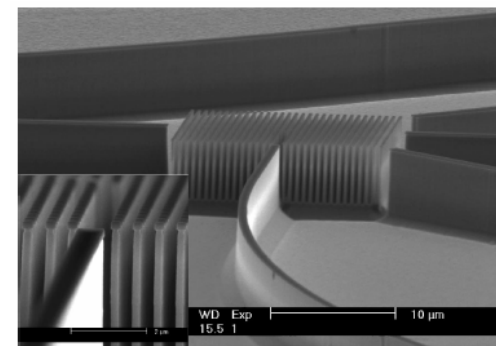
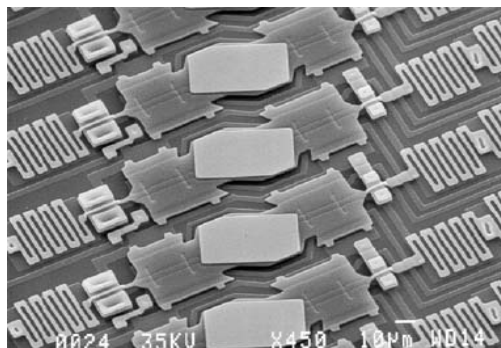
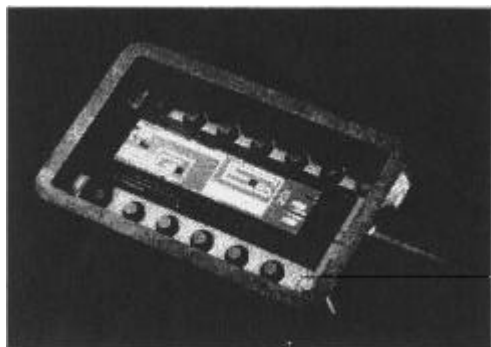
- **Photonics: the next “killer” application (X100/10yrs)**



**... but based on unsustainable manufacturing processes**

## • Si microphotronics

- Microphotronics based on Si and Si-compatible processes
- Bringing the astronomical Si processing infrastructure and know-how to photonics
- Si itself is an excellent optical material with high transparency in the infra-red region and chemical/mechanical strength
- But lack of optical activity limit its use to passive devices



... because if you can do it with silicon, you **MUST** do it with silicon

# Hydrogenated amorphous Si and SiC (a-Si:H and a-Si:H:C)

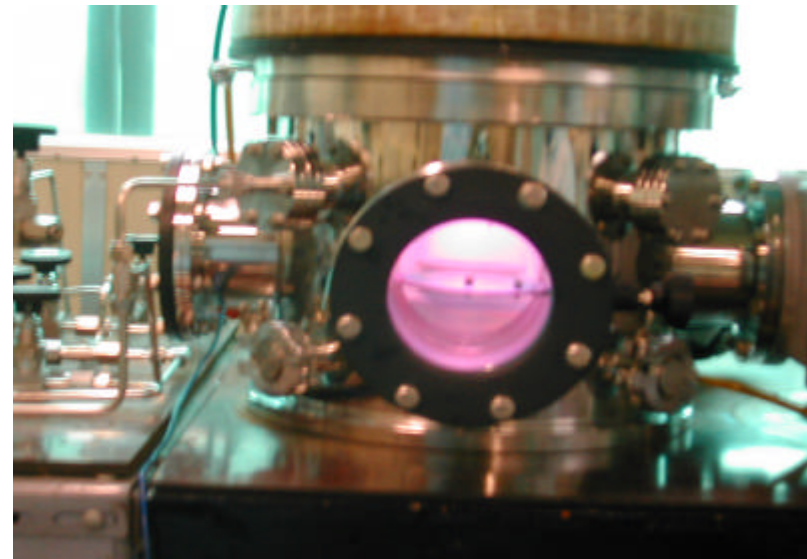
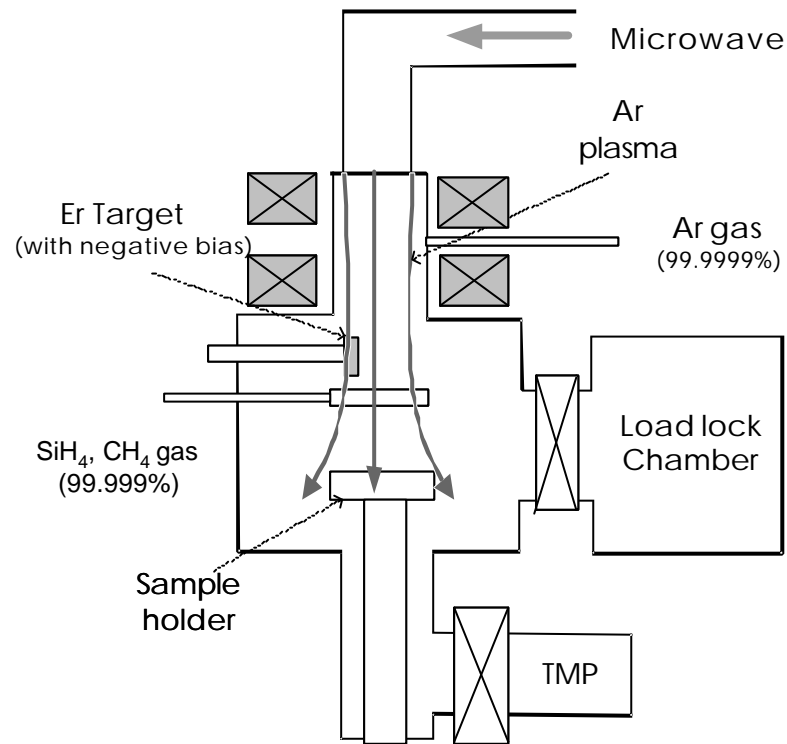
- **a-Si:H: Low-cost alternative to crystalline Si**

- Large area, low-cost deposition on optical substrates (quartz, glass)
- Highly transparent in the infra-red region ( $< 3\text{dB cm}^{-1}$  Peuzin, Opt. Comm 1984)
- Variable refractive index for added freedom in design
- Very smooth surface and compatibility with the standard Si processes for smooth sidewalls
- Low-temperature deposition allows photonics fabrication to be a back-end process

- **a-Si:H:C**

- Isovalent impurity with Si: for small carbon concentration ( $< 40\%$  of Si), very similar to a-Si:H in terms of material and electronic structure
- Mature material – used frequently in high-efficiency solar cells
- Larger bandgap than a-Si:H
- Carbon demonstrated to be an activating ligand for Er (Michel, PRB 1991)

# Deposition of the Er-doped a-Si:H:C



- **ECR-PECVD of SiH<sub>4</sub> and CH<sub>4</sub> with concurrent sputtering of RE**

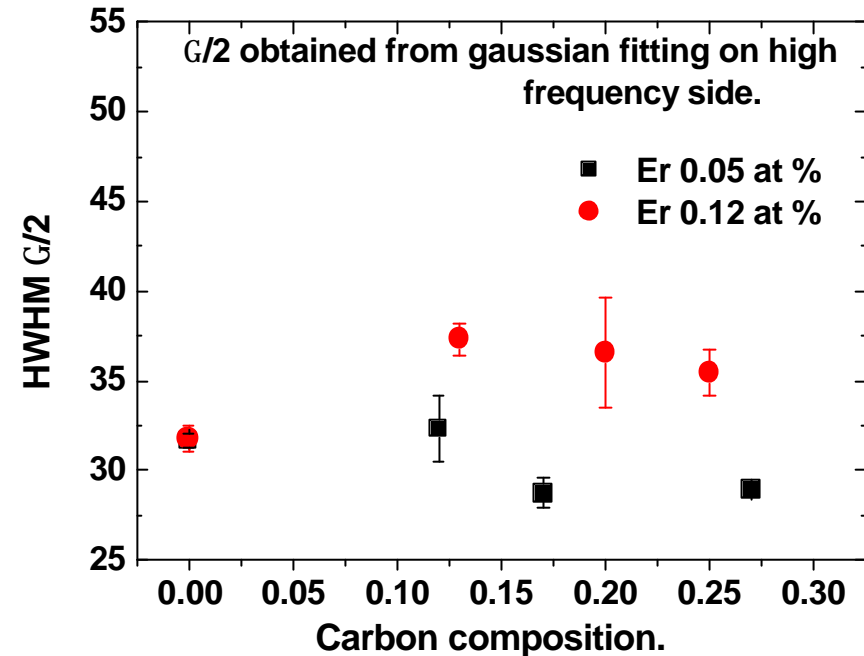
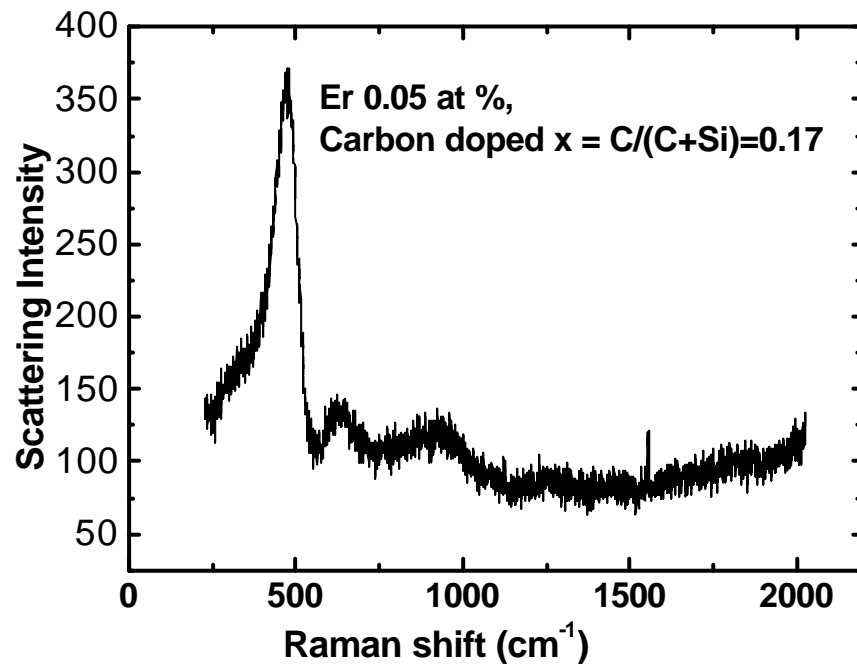
- No metalorganic precursor needed → no organic contamination
- Substrate: Corning 1737 glass
- Base pressure:  $1 \times 10^{-7}$  torr
- Microwave power: 400W

Dep. Pressure:  $1 \times 10^{-4}$  torr

Dep. Temp: 250

# Results

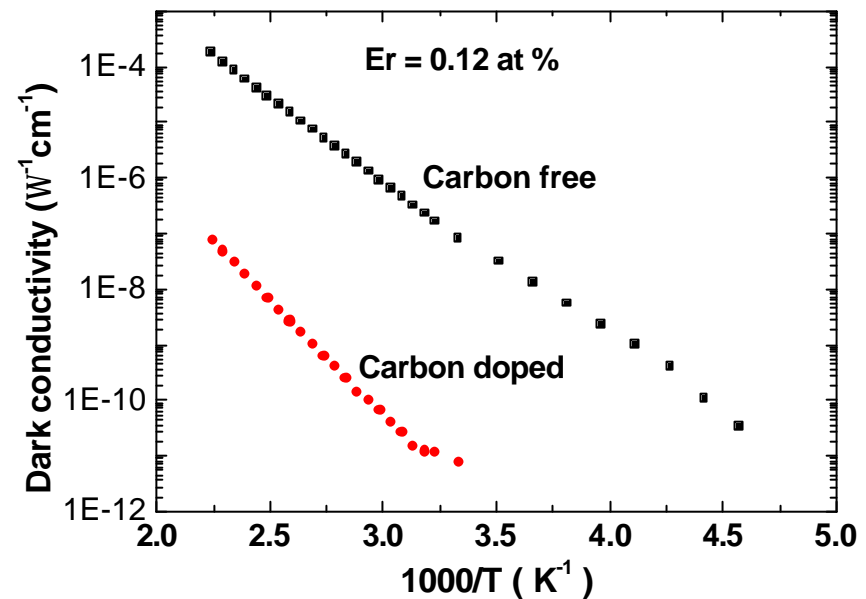
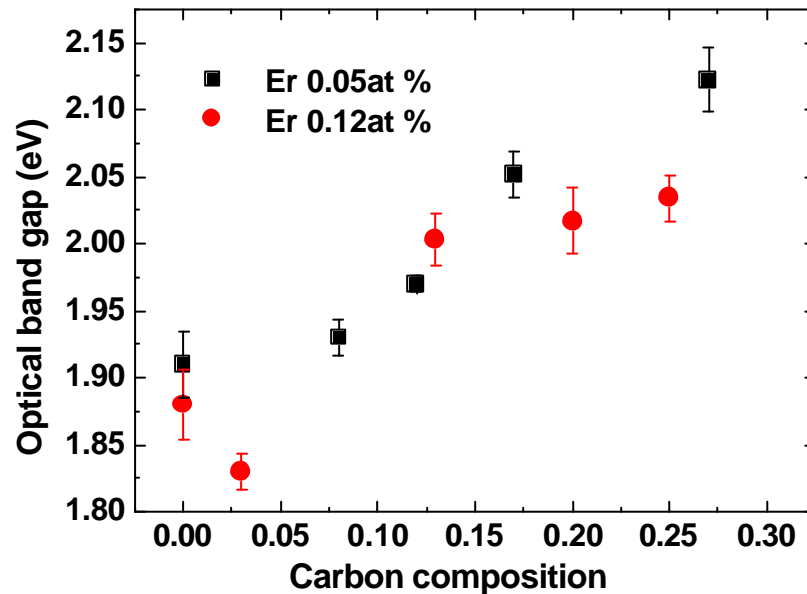
- Material structure



- Raman spectra of Er-doped a-Si:H:C

- Strong Si-Si bond ( $300\text{-}600\text{ cm}^{-1}$ ), weak Si-C bond ( $600\text{-}1000\text{ cm}^{-1}$ ), but no C-C bond ( $1300\text{-}1600\text{ cm}^{-1}$ )  $\rightarrow$  No carbon clustering, with carbon mostly in  $\text{CH}_2$  and  $\text{CH}_3$  form
- $\Gamma/2$  value of  $32\text{ cm}^{-1}$   $\rightarrow$  good short range order with little bond angle distortion
- Carbon co-doping does not increase  $\Gamma/2$  value  $\rightarrow$  carbon does not induce structural disorder

- **Electronic structure**

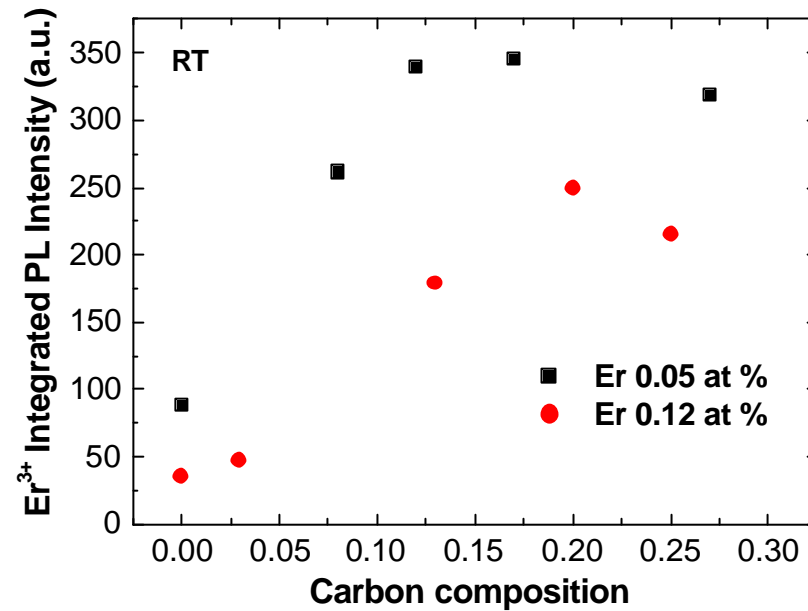
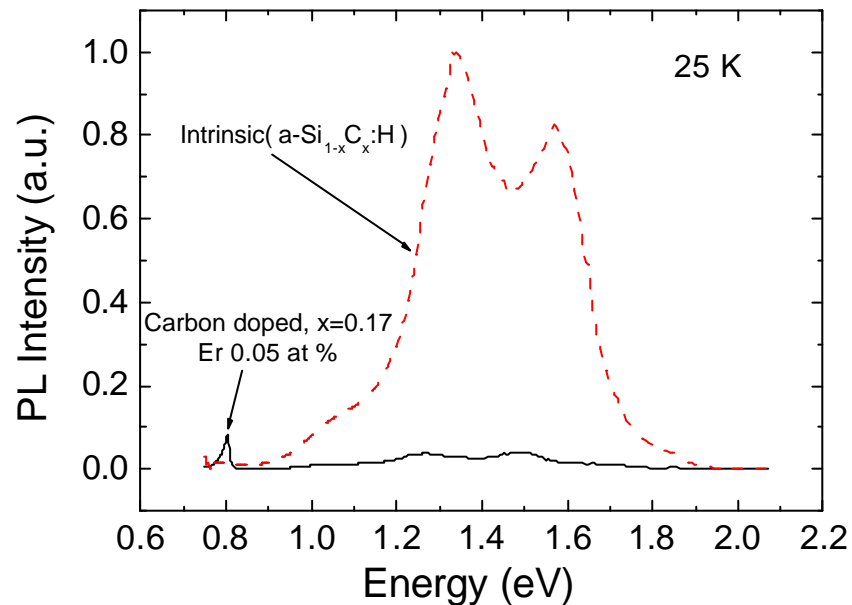


- **Effect of carbon co-doping:**

- Carbon co-doping Increases the optical bandgap (Tauc gap) irrespective of Er concentration
- Carbon co-doping strongly reduces the dark conductivity while retaining semiconducting behavior, consistent with the larger bandgap



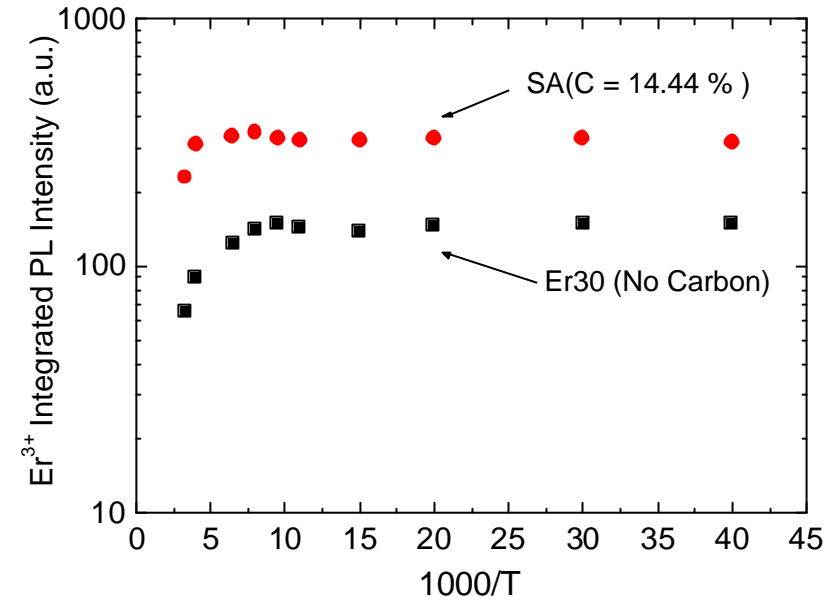
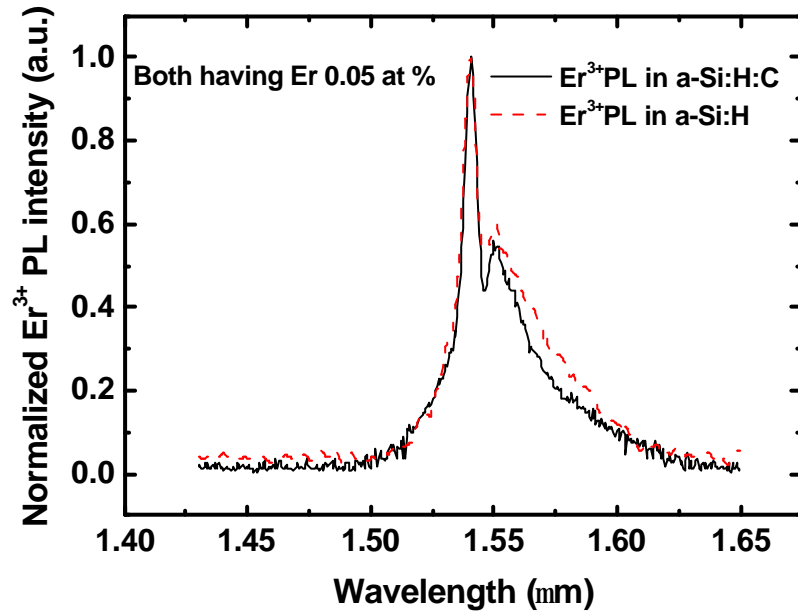
- **Photoluminescence**



- **Photoluminescence using 477 nm line of Ar laser**

- Er doping nearly completely quenches the intrinsic a-Si:H:C luminescence (peaks in the a-Si:H:C PL are optical artifacts due to multiple reflection in the film)
- Er PL intensity is much weaker than the a-Si:H:C PL intensity
- Up to 4fold increase in the RT Er PL intensity by C co-doping!

- Cont'd



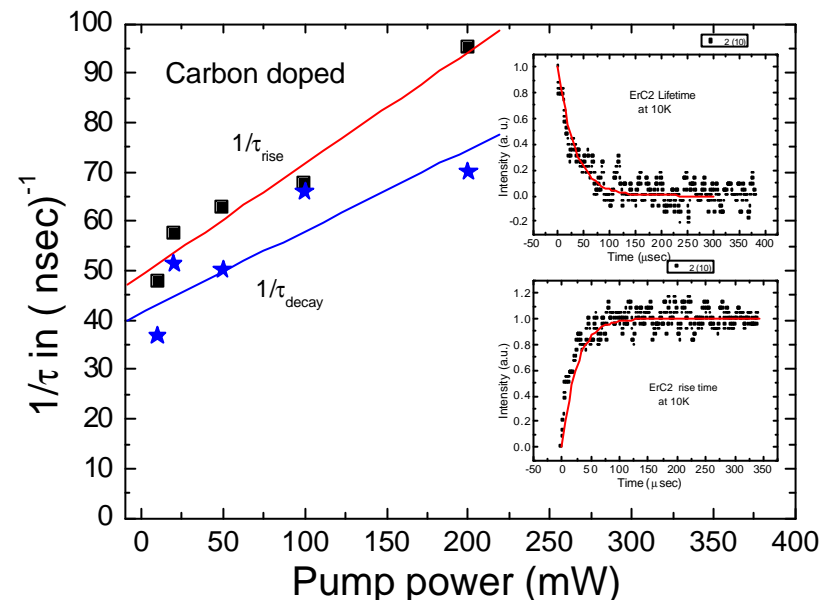
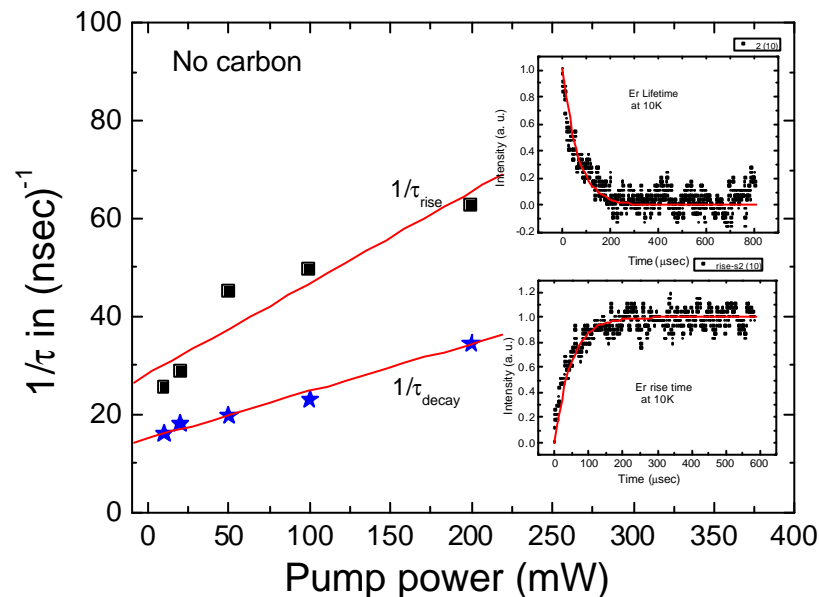
- Carbon co-doping

- Has only small effect on  $\text{Er}^{3+}$  PL spectrum
- Reduces temperature quenching from 50 to 30 %

# Discussion

- **Effects of carbon on Er<sup>3+</sup> luminescence**

- Getting of carbon by Er: No change of Er<sup>3+</sup> PL spectra → likely not critical
- “Activation” of Er by carbon: 2 fold PL increase upon 100 fold increase in C → likely not critical
- Matrix – Er interaction?

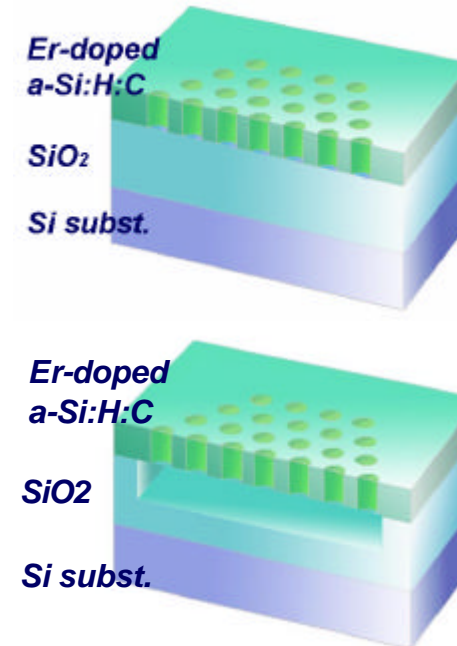
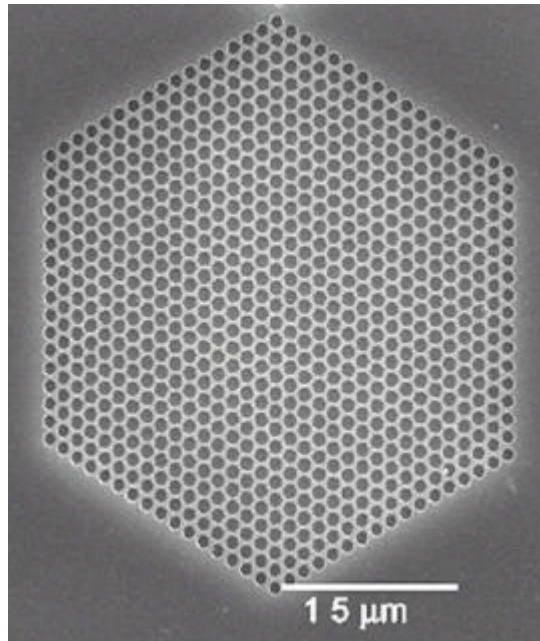


- **Time resolved measurements of luminescence rise and decay**

- Difference in slope gives effective excitation cross section:  $1 \pm 0.5 \times 10^{-15} \text{ cm}^{-2}$  for both carbon-free and carbon doped film
- The most important effect of carbon co-doping is suppression of temperature quenching

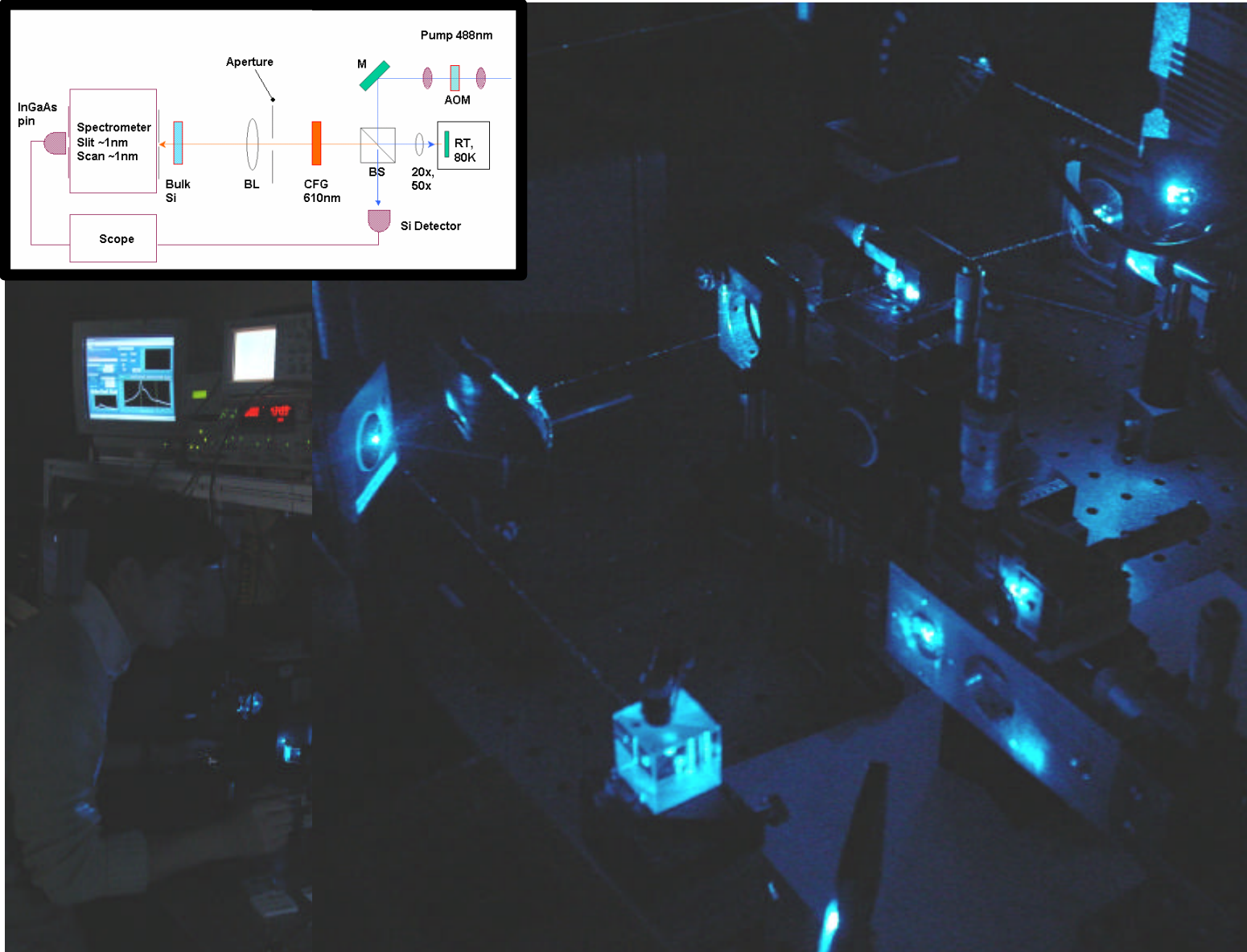
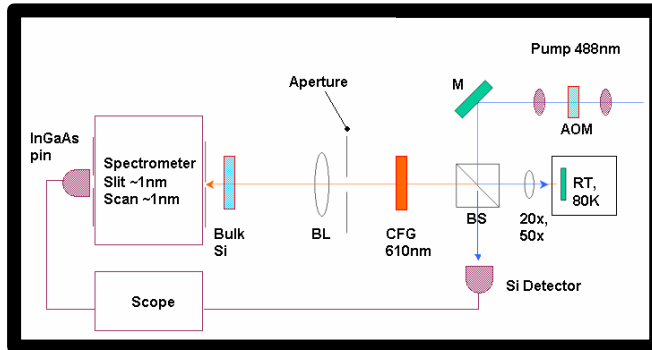
# Microphotronics using a-Si:H:C:Er

- **2D slab photonic crystal using Er-doped a-Si:H:C**
  - a-Si:H:C: sufficient refractive index contrast for photonic crystal
  - a-Si:H:C:Er → active, Si-based photonic crystal!

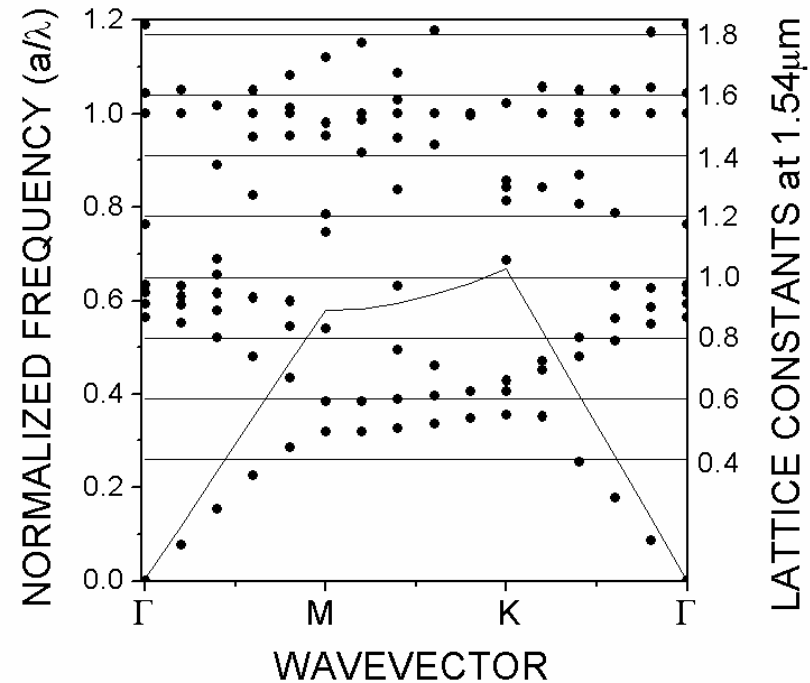
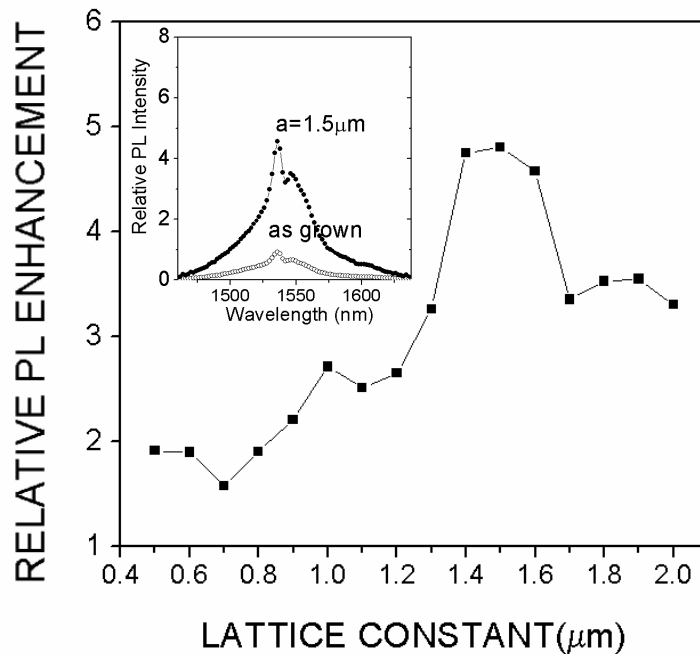


- 0.5 μm thick Er-doped a-Si:H:C on 5 μm thick thermal oxide
- Triangular lattice of holes with  $r/a$  fixed at 0.3
- Au + PMMA → E-beam litho → Ion milling → RIE → plasma ashing + mask removal
- Free-standing structure formed by BOE etching of SiO<sub>2</sub> layer

# Experimental setup

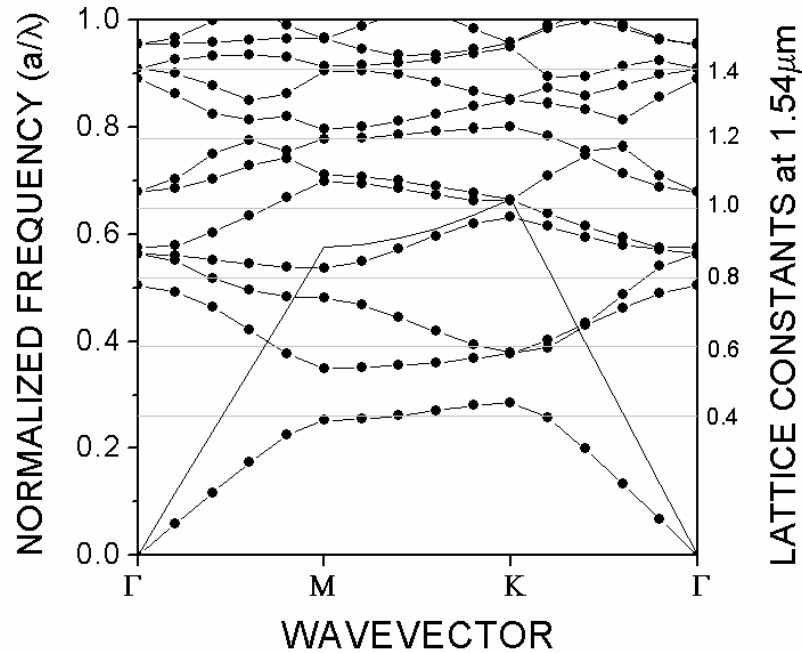
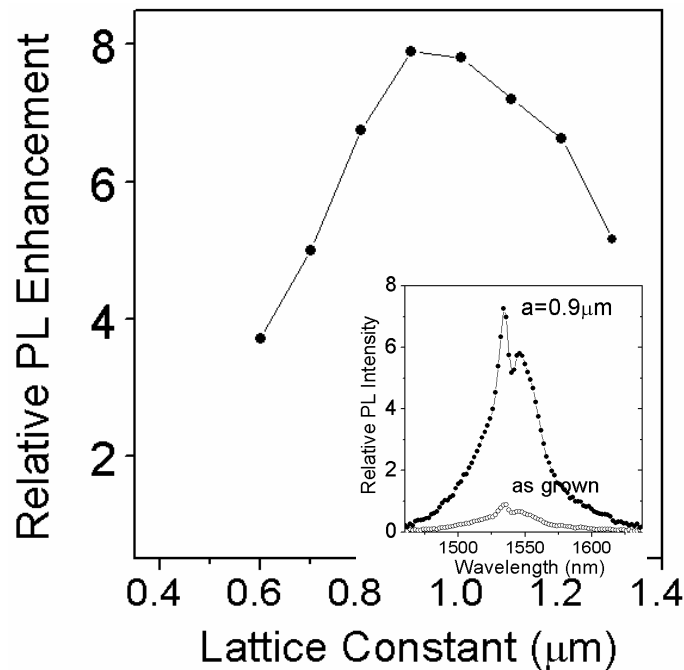


- **Enhanced light extraction from photonic crystal: oxide clad structure**
  - Strong coupling of in-plane radiation with vacuum mode



- 3D FDTD calculation of photonic bands
- 5 fold relative PL enhancement near normalized frequency of 1 at room temperature

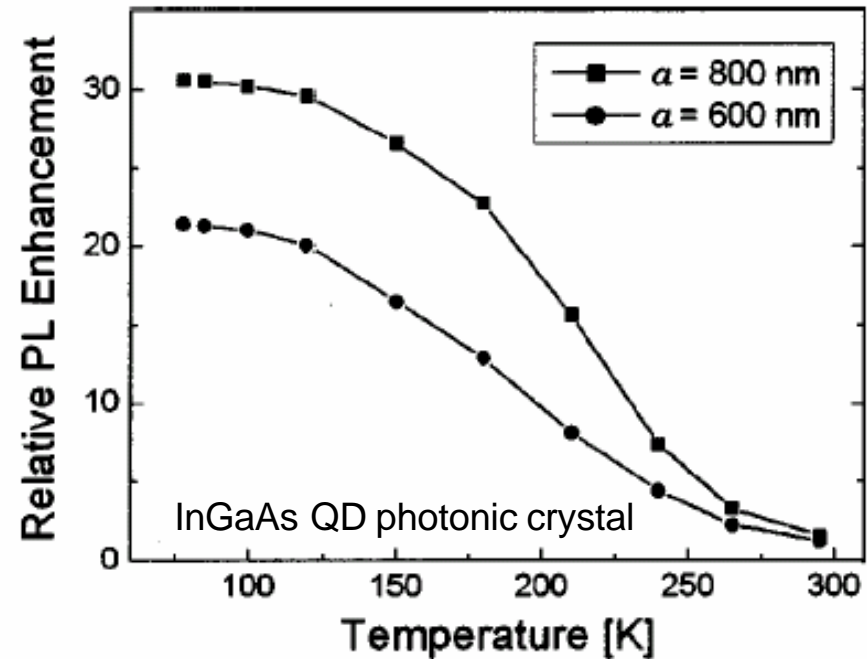
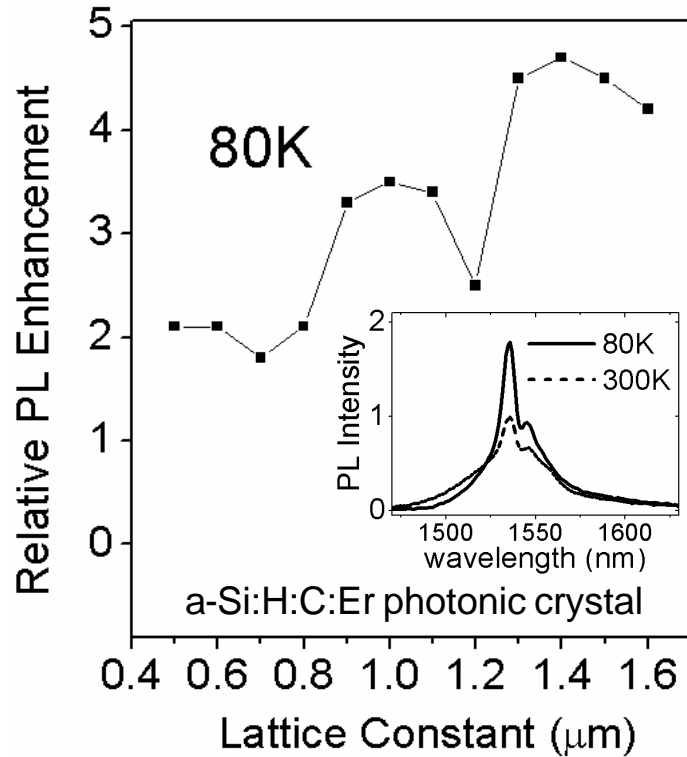
- **Enhanced light extraction from photonic crystal: free-standing structure**
  - Stronger coupling of in-plane radiation with vacuum mode



- 3D FDTD calculation of photonic bands
- Full bandgap for TE-like mode observed
- 8 fold relative PL enhancement near normalized frequency of 0.6

- **Advantage of using Er-doped a-Si:H:C**

- Insensitivity to temperature

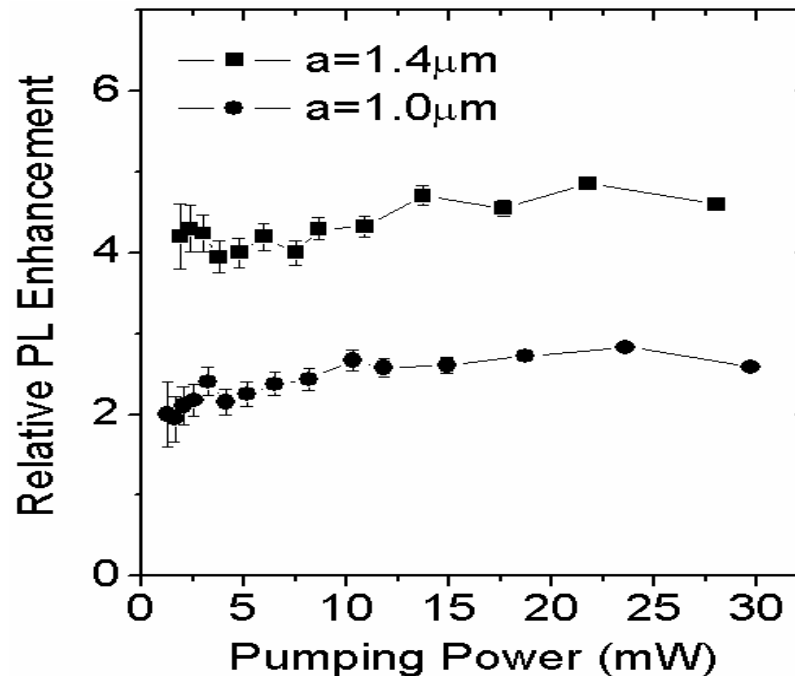


- Near complete independence of extraction efficiency on temperature, quite unlike that observed for III-V compound semiconductor-based photonic crystals
- Slight hump near 0.6 due to change in PL shape



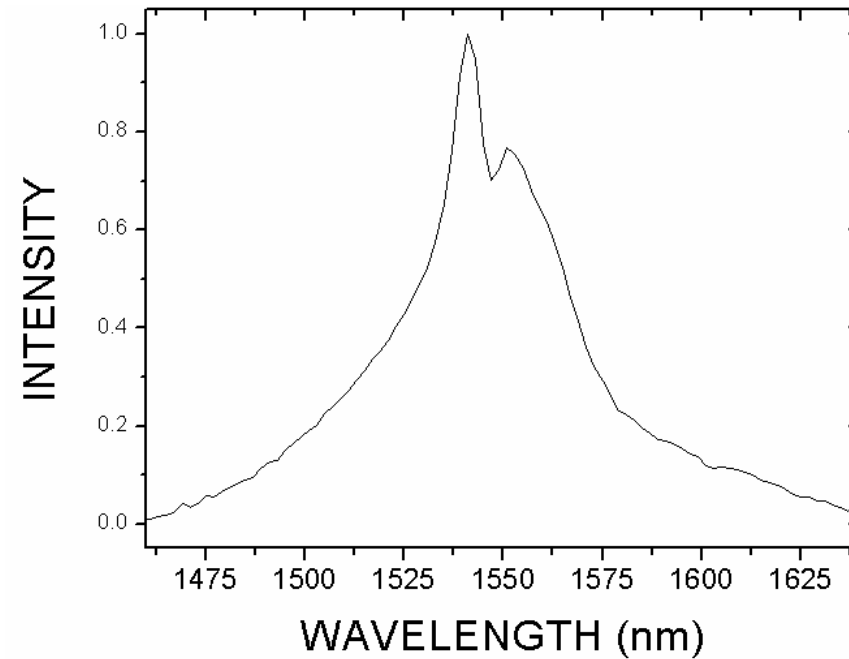
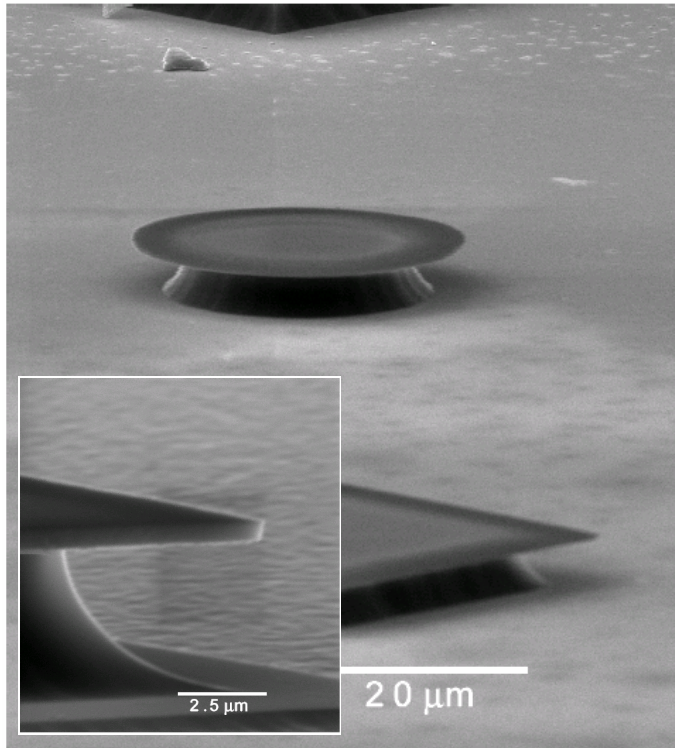
- **Advantage of using Er-doped a-Si:H:C**

- Insensitivity to pump power



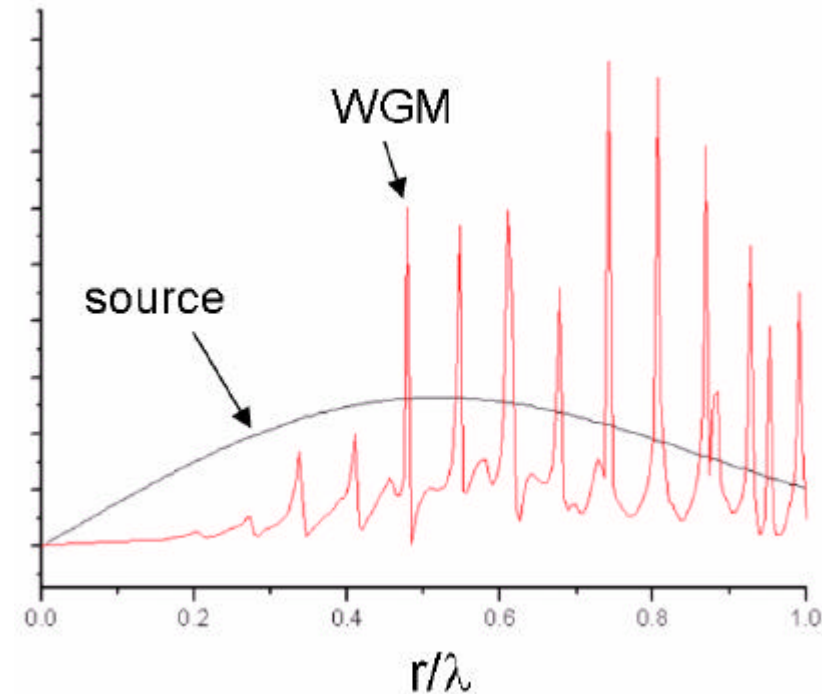
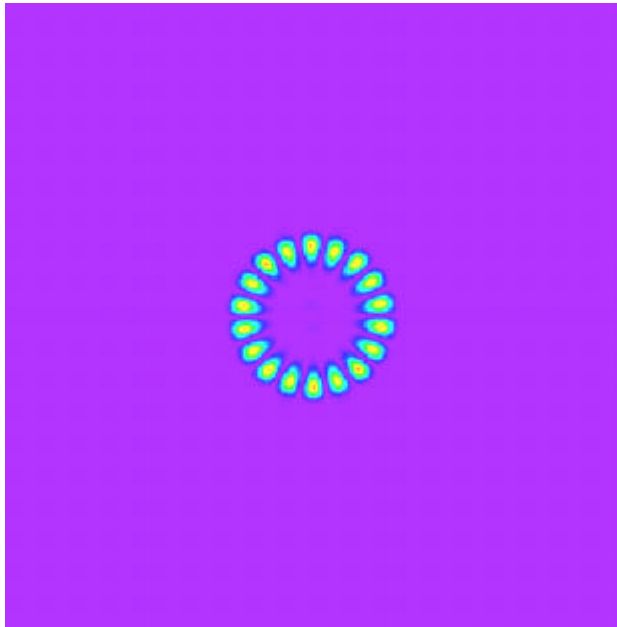
- Near independence of the extraction efficiency on the pump power, quite unlike that observed for III-V compound semiconductor-based photonic crystals
- Surface recombination nearly negligible for  $\text{Er}^{3+}$  luminescence from a-Si:H:C → consequence of an atomic transition excited via trapped carriers

- **Microdisk resonator based on Er-doped a-Si:H:C**



- Microdisks with smooth wall and edges formed by optical litho and BOE etch
- Er<sup>3+</sup> optical activity unaffected by lithography and etching

- Expected modes in fabricated microdisks



- Whispering gallery modes for 10  $\mu\text{m}$  microdisk: mode spacing of only 6 nm  $\rightarrow$  too low S/N ratio to observe mode vertically
- For a microdisk with moderate Q of  $10^5$ , the lifetime of WGM is 80 ps ( $T = Q/(2\pi(\lambda/c))$ )  $\rightarrow$  it will travel for 1 cm!

# Conclusion

- **Er-doped a-Si:H:C demonstrated as a viable material for active, Si based microphotronics**
- **Carbon co-doping results in 4-fold increase in RT Er<sup>3+</sup> PL intensity without adversely affecting material or electronic structure of the material.**
- **2D slab photonic crystal based on Er-doped a-Si:H:C demonstrated**
  - **Up to 8-fold increase in the relative PL extraction efficiency**
  - **Near complete insensitivity to the environmental factors**
- **Microdisks based on Er-doped a-Si:H:C demonstrated**
  - **Process-insensitive Er<sup>3+</sup> PL demonstrated**

## Acknowledgment

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